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Combined strain rate, mesh size and calibration test influence on structural failure: miniature ship grounding test

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Abstract

This paper investigates the influence of the strain rate and element size on the fracture of steel plates presented in marine structures undergoing grounding when modeled by the finite element method. Three different experimental tests (tensile, tearing and perforation tests), performed at different velocities, are used in the study. These tests are conceived to cover diverse rupture modes related mainly to crack initiation and propagation. Plastic strain- and stress state-based failure criteria were calibrated from these experiments. Sensitivity to strain rate and mesh size were also evaluated. A ship grounding event was reproduced experimentally in 1:100 reduced scale. This experiment was FE modeled to evaluate the performance of the aforementioned failure criteria. The results verified the significant influence of the strain rate and stress state on the failure strain. In addition, it is shown that the sole use of tensile tests is not enough to properly characterize the material failure observed in the structural collapse mode of the miniature ship bottom structure. This collapse is shown to be a combination of stretching and tearing of structural members.

keywords: material failure, strain rate sensitivity, tensile tests, tearing tests, perforation tests

1. Introduction

During the last decades, several researchers investigated collision accidents of marine structures using numerical methods. The finite element method showed to be one of the most reliable tools to predict the acting forces and damaged configuration of marine structures when subjected to collision/grounding tests.

In the last few years, the rapid increase of computational processing power allowed the consideration of diverse non-linearities so that the output results became more and more accurate. Clearly, it is valuable to discern what aspects are more or less relevant in an analysis so one can retrieve accurate output results.

Among the main aspects that affects the accuracy of a FE analysis one can list:

- Shell elements: In view of the huge dimensions of marine structures when compared to the plate thickness used in shipbuilding, the use of shell elements became mandatory to reduce the number of elements so avoiding long computational times. This is true despite of the fact that shell elements incorporate rotational degrees of freedom and local systems, which are computationally difficult to parametrize/update, and are inadequate to model large longitudinal and transverse shear strains (Flores, 2016). Even so, shell finite elements are still considered as the standard element to model marine structures subjected to collision (Calle and Alves, 2015; Liu et al., 2018).
- Strain rate sensitivity on flow stress: Taken into account the dynamic character of ship collision/grounding events, the strain rate sensitivity on flow stress of shipbuilding steels is being included in recent numerical-experimental works when material undergoes plastic deformation at high loading rates (Choung et al., 2010; Choung et al., 2013; Storheim and Amdahl, 2017; Paik et al., 2017). To do it, some well-known constitutive material models, such as Cowper-Symonds (1957) and Johnson-Cook (1985), are being used so making evident the strong influence of the strain rate on the plastic hardening of steels.
- Material fracture: It is widely known that the definition of the parameters of the failure criteria is the most important key point for a correct prediction of a realistic structural collapse mode and an acceptable quantification of the impact energy absorption in the FE modeling of ship collision and grounding events (Calle and Alves, 2015). The equivalent plastic failure strain is the most common

material fracture criterion employed in the analyses of ship collision and grounding events. In this work, the equivalent plastic strain at failure is also denoted “failure strain” as widely termed by diverse works in marine engineering (Ehlers and Varsta, 2009; Choung et al., 2012; Storheim et al., 2015; Marinatos and Samuelides, 2015). The failure strain is usually evaluated experimentally via uniaxial tensile tests. However, this value depends on many factors and exhibits a scatter as large as 10% to 70%, according to the literature review made by Calle and Alves (2015). For instance, early works on FE modeling of ship collision already considered the numerical influence of the shell element size on the failure strain according to Barba’s law. So, uniaxial tensile tests needed to be FE modeled at different element sizes to calibrate the failure strain (Peschmann and Kulzep, 2000; Yamada et al., 2005; Ehlers, 2010; Hogström, 2012). However, some other physical factors are more difficult to be considered into the material failure criteria as commented below.

- Strain rate sensitivity on material fracture: this is a challenging aspect to be considered and implemented in FE codes and in failure criteria (Paik, 2007a; Kim et al., 2016; Paik et al., 2017, Storheim and Amdahl, 2017). Jones (1989) proposed an inverse form of the Cowper-Symonds formulation that reproduces the experimental decrease observed in the failure strain when strain rate rises (Paik and Pedersen, 1996; Yamada et al., 2005; Paik, 2007b). However, the parameters for this inverse formulation differ to that for the traditional Cowper-Symonds formulation for mild steels ($C = 40.4 \text{ s}^{-1}$, $p = 5$) (Paik and Thayamballi, 2003; Paik, 2007a; Paik, 2007b) so requiring, anyway, a particular mechanical characterization procedure. The influence of the strain rate on the material fracture varies from one material to another (Paik et al., 2017) and, depending on the analyzed strain rate range, also requires the consideration of temperature effects by the model (Roth and Mohr, 2014). Additionally, it is not completely clear how the strain rate influences the structural response of large marine structures subjected to collision. For these reasons, most of the researches assume low structural influence of strain rate on material fracture during ship collision events or, basically, assume a lower equivalent plastic strain at failure due to dynamic influence (Endo et al., 2002; Yagi et al., 2009; AbuBakar and Dow, 2013; Bela et al., 2017; Farinha et al., 2018; among others).

- Crack propagation: FE modeling of material fracture is further complicated when it is induced by crack propagation. Most of the material fracture criteria are conceived for FE modeling of the crack initiation process, so they work accurately when modeling the first element removal. After that, the crack shape remains unrealistic and unrealistic levels of energy to propagate it are obtained. This leads to modeling errors (Calle and Alves, 2015). Uniaxial tensile tests are widely used for mechanical characterization of the materials including fracture evaluation. In this kind of test, the crack initiation process prevails in the rupture process once the crack is generated in the middle section of the sample and the complete rupture given by the propagation of the crack is abrupt. Thus, some authors perform experimental and numerical studies focused on crack propagation (Simonsen and Törnqvist, 2004; Kõrgesaar and Romanoff, 2013). As a result, Simonsen and Törnqvist (2004) obtained an increase of 40% in the equivalent strain at failure when compared to that calibrated by tensile test. However, so far, no unified approach to cover both initiation and propagation of fracture exists.
- Stress state on material fracture: material failure criteria based on the equivalent plastic strain (failure strain criteria) are inaccurate when the FE modeled structure is subjected to stress states different of the one presented in uniaxial tensile tests, e.g. biaxial tension or shear loadings. Some authors justified the use of a failure strain criterion by the low variation of triaxialities in fracture during a ship collision/grounding event or by making use of an average plastic strain value to include both crack initiation and propagation phenomena (Calle et al., 2017a). Some authors developed material failure criteria to model collision of marine structures that includes the stress states into its formulations (Lehmann and Yu, 1998; Kitamura and Endo, 2000; Törnqvist, 2003; Kõrgesaar and Romanoff, 2014; Kõrgesaar, 2019) so obtaining accurate modeling results.

Together with these aspects, it is worth mentioning that ship grounding is a complex event in which the major part of energy dissipation occurs in inelastic straining (Nguyen et al., 2012). Two possible scenarios of ship grounding can be considered (Paik and Thayamballi, 2003): the first, called properly “grounding” or “powered grounding”, results in a raking damage of the ship bottom structure induced by a horizontal slide over a seabed obstacle. The second, called “stranding”, takes place when the ship bottom structure at standstill is vertically

pressed by an obstacle due to pitch and/or heave ship movements. In general, real grounding accidents combine these two scenarios, with penetration and tearing of the ship bottom structure. As a consequence, structural members are subjected mainly to lateral-edge (tearing or cutting) and in-plane (membrane) forces, so inducing high and low levels of energy dissipation respectively. According to Nguyen et al. (2011), this energy dissipation is the key aspect for modeling. Besides, the contour geometry of seabed obstacles (rock, reef or shoal) can also affect this energy dissipation. On the one hand, rock obstacles easily perforate outer plates so inducing local damages; on the other hand, large shoal obstacles may deform larger parts of the bottom structure (Alsos and Amdahl, 2007).

Having in mind this scenario, this paper presents an experimental investigation about the influence of strain rate on the material fracture of a mild steel plate aiming the finite element modeling of a ship-grounding event. A miniature ship grounding experiment was also performed to validate the numerical results. Three different mechanical tests (uniaxial tensile, perforation and out-of-plane tearing) are used to calibrate the failure strain at different strain rates, intended to recreate different rupturing modes observed in ship collision and grounding, particularly that related to crack initiation (tensile and perforation) and propagation (tearing).

2. Base material strain hardening

2.1 Material and experiments

Uniaxial tensile tests were performed in a cold-rolled SAE 1008 carbon steel sheet with 0.25 mm thickness subjected to quasi-static and dynamic conditions.

In order to evaluate the mechanical behavior in quasi-static conditions, standard dog-bone specimens were subjected to uniaxial tensile test. The dimensions of the reduced section of the specimen are 50 mm length × 10 mm width. The test velocity was set in 0.0025 mm/s so inducing an average strain rate of $4.1 \times 10^{-5} \text{ s}^{-1}$.

The strain data was acquired by Digital Image Correlation (DIC) technique employing a virtual mesh on the sample surface with uniform 0.5 mm side square elements. A maximum plastic strain at failure of 0.5822 was measured in the middle of the ruptured specimen section at a strain rate of $3.8 \times 10^{-4} \text{ s}^{-1}$.

Given that the mechanical strength of carbon steels increases at higher strain rates, small dog-bone specimens were subjected to uniaxial tensile tests to evaluate the strain rate sensitivity of the material. Tensile tests at

different test velocities were performed in an earlier work (Calle et al., 2017b) using small tensile specimens based on a Hopkinson bar machine tensile samples designed by Verleysen et al. (2008). The dimensions of the reduced section of the small specimen are 5 mm length $\times$ 4 mm width, and the distance between the sample shoulders is 9 mm.

For all tensile tests, the true stress-strain curves are obtained from engineering stress-strain curves. At the same time, the engineering stress-strain curves are obtained from the force-displacement responses of the tensile test experiments as follows:

$$\sigma = (1 + e) S \quad (1)$$

$$\varepsilon = \ln(1 + e) \quad (2)$$

$$S = F/A_0 \quad (3)$$

$$e = \delta/L_0 \quad (4)$$

where σ and ε are the true stress and true strain respectively, S and e are the engineering stress and engineering strain respectively, F and δ are the force and displacement acquired from the testing machine, L_0 and A_0 are the initial length and area of the tensile samples.

The instantaneous strain rate is defined as the rate at which the strain occurs and is given by

$$\dot{\varepsilon} = d\varepsilon/dt \quad (5)$$

Particularly for the analysis of the increment of the true stress level at dynamic conditions, Eq. (12), an average strain rate is considered by averaging the strain rates in between yield point and below the necking point as

$$\bar{\varepsilon} = \frac{1}{\Delta t} \int \dot{\varepsilon} dt \quad (6)$$

2.2 Strain rate-dependent plasticity model

The material is assumed as isotropic and the elasto-plastic constitutive law follows the J_2 -flow theory, i.e., the Von Mises yield surface with isotropic hardening and associated flow rule as

$$\Theta(\bar{\sigma}, \sigma_f) = \bar{\sigma} - \sigma_f(\varepsilon, \dot{\varepsilon}) = 0 \quad (7)$$

$$\bar{\sigma} = \sqrt{3J_2} \quad (8)$$

$$J_2 = \frac{1}{6}[(\sigma_1 - \sigma_2)^2 + (\sigma_2 - \sigma_3)^2 + (\sigma_3 - \sigma_1)^2] \quad (9)$$

where $\bar{\sigma}$ is the Von Mises equivalent stress, the flow stress $\sigma_f(\varepsilon, \dot{\varepsilon})$ is assumed as a function of the equivalent plastic strain ε and the equivalent plastic strain rate $\dot{\varepsilon}$.

The increment of equivalent plastic strain is given as a function of the principal strain components as

$$d\varepsilon = \sqrt{\frac{2}{3} [d\varepsilon_1^2 + d\varepsilon_2^2 + d\varepsilon_3^2]} \quad (10)$$

A power law constitutive model to reproduce the true stress strain behavior of the material, at quasi-static loading conditions, is given as follows

$$\sigma_0 = \begin{cases} \sigma_y & \varepsilon \leq \varepsilon_L \\ K\varepsilon^n & \varepsilon > \varepsilon_L \end{cases} \quad (11)$$

where σ_0 is the true stress at quasi-static conditions, ε is the equivalent plastic strain, σ_y is the yield stress, K and n are the material parameters and $\varepsilon_L = (\sigma_y/K)^{1/n}$ to ensure curve continuity. Parameters K and n are adjusted using a log-log approach, considering the points above the yield point and region below the necking point, and listed in Table 1.

The increment on the true stress levels due to the increment on the strain rate is widely evaluated at a reference plastic strain value commonly set between 5% and 20%. Notwithstanding, in several cases, it results in a poor fitting at higher strain rates due to the gradual material thermal softening. Thermal softening occurs by the adiabatic heating of the material when strained at high velocities (Mason et al., 1994; Kapoor and Nemat-Nasser, 1998).

Aiming at a better fit, even at high strain rates, the strain rate sensitivity constitutive model given by Alves (2000) was adopted to introduce the strain rate sensitivity to the flow stress as follows

$$\sigma_f = \sigma_0 + \sigma_y \left(\frac{\dot{\varepsilon}}{C} \right)^{1/p} \quad (12)$$

where σ_f is the flow stress (or true stress at dynamic/general conditions), σ_0 is true stress at quasi-static conditions, Eq. (11), σ_y is the yield stress, $\dot{\varepsilon}$ is the equivalent strain rate, C and p are material parameters obtained by data log-log fitting. The material parameters, obtained at 10% plastic strain, are presented in Table 1.

Fig. 1. Experimental and fitted true stress-strain curves at different strain rates.

Table 1. Material parameters.

σ_y	226.9 MPa
K	626.9 MPa
n	0.2077
C	1346.9 s ⁻¹
p	6.4671

Next, this material model was implemented in the commercial code Abaqus/CAE 2016 – explicit analysis. Shell elements (homogeneous type) with five integration points are considered in all FE models. The coupled Power law plus Alves model (Eq. 12) is used to describe the plastic strain hardening and strain rate sensitivity by listing the “yield stress” as a function of the “plastic strain” and “rate” (strain rate). The failure strain is modeled by using the “ductile damage” parameter. To reduce the processing time, “semi-automatic mass scaling” technique is used in the numerical modeling of all mechanical tests.

3. Base material failure

3.1 Experiments on uniaxial tensile loading

Failure strain for plate metals are commonly obtained using standard uniaxial tensile specimens. However, the long length of standard dog-bone specimens does not allow achieving high strain rates (Verleysen et al., 2008). Tensile tests in small specimens (section 2) are performed here to calibrate the strains at failure.

1 The tensile tests in small specimens are FE modeled to reproduce the material rupture at different test
 2 velocities. Uniform 0.5×0.5 mm shell elements are considered for the FE model. An iterative calibration
 3 procedure was performed in order to equate numerical and experimental elongation-to-rupture responses at
 4 each test velocity.

5 Given the relatively large element size used in the FE modeling of small specimens when compared with the
 6 specimen dimension, the after necking force response in the tensile test become infeasible to be FE modeled
 7 accurately. Moreover, at higher test velocities, numerical and experimental force-elongation curves diverge
 8 moderately at large plastic strains because of the material softening, as commented previously in section 2.
 9 For these reasons, material failure was calibrated using as a reference the equivalence in absorbed energy
 10 before material fracture, i.e., the area under the force-displacement curves.

11 The influence of mesh size on the failure strain is evaluated with the quasi-static tensile test with standard
 12 dog-bone specimen (section 2). This test was FE modeled considering different mesh sizes with ratios of
 13 equivalent length over thickness (L_e/t) in a range from 2.0 to 20 (0.5×0.5 , 0.625×0.625 , 1.0×1.0 , $1.667 \times$
 14 1.667 , 2.5×2.5 , 5×3.333 and 5×5 mm). These models are presented in Appendix A.1. For rectangular-shaped
 15 shell elements with dimensions $L_1 \times L_2$, the equivalent length is defined as $L_e = \sqrt{L_1 \cdot L_2}$. An elongation at
 16 rupture of 18.3 mm was set as a common reference to calibrate the failure strains for different mesh sizes. It
 17 corresponds to the elongation at rupture obtained numerically considering 0.5822 failure strain as
 18 experimentally measured using DIC technique.

Fig. 2. Experiment setup and force responses obtained from uniaxial tensile tests in small specimen at different velocities.

3.2 Experiments on perforation loading

Perforation tests in $\varnothing 80$ mm circular plates were conducted at different indentation velocities. Circular specimens were peripherally clamped in a cylindrical holed support using clamps with serrated surfaces in order to prevent sliding of the plate boundaries. A rigid cylindrical indenter with 90° cone head spherically smoothed by a 3.4 mm radius was used.

The perforation tests were performed in three different experimental setups: universal screw-driven test machine, all-electric dynamic testing machine and a low-energy drop weight machine. In the drop weight machine, a falling mass of 0.76 kg was used and its velocity, $\dot{u}(t)$, was acquired by a laser Doppler vibrometer. Then, the mass displacement and acceleration were obtained by integrating and deriving the velocity signal such as $u = \int \dot{u}(t) dt$ and $\ddot{u} = d\dot{u}(t)/dt$ respectively and the force by multiplying acceleration by the falling mass, $F(t) = m \ddot{u}(t)$.

Fig. 3 shows the force-displacement responses obtained in the perforation tests. Higher force slopes are observed at higher test velocities. In contrast, force peaks came down when the test speed was increased. Moreover, diverse structural responses are observed after force peak is achieved.

Fig. 3. Experiment setup and force responses obtained from perforation tests at different velocities.

In order to calibrate the plastic strain at failure in the perforation test, FE modeling of the perforation tests at different velocities were performed using uniform 0.5×0.5 mm shell elements. Given that no slipping was

detected between the circular support and the plates in the experiments, the FE model of the circular plate was fully constrained in its perimeter. The plastic strain at failure is iteratively calibrated so to obtain numerically a maximum peak force equivalent to that obtained experimentally for each test velocity. In order to evaluate the influence of the mesh size on the material failure when modeling a perforation test, FE modeling of the perforation test using different mesh sizes of shell elements are performed. The perforation test response at 0.0025 mm/s was used as reference, which resulted in a peak force of 1.65 kN. Shell element sizes with equivalent length over thickness ratios (L_e/t) ranging from 2 to 20 (0.5×0.5 , 1×1 , 2×2 , 3.333×3.333 , 4×4 and 5×5 mm) were employed. The geometry of these models are shown in Appendix A.2. Given the relevant influence of friction on the stress state and ductile fracture of punched plates (Wiśniewski and Kołakowski, 2003; Lee et al., 2004), FE models of perforation tests considering friction coefficients between 0.1 and 0.3 were developed. Independently of the test speed, all perforation experiments generated a fracture in triangular pattern (projection of three cracks from the central area). When FE modeling the perforation test using a mesh size of 0.5×0.5 mm, a friction coefficient of 0.1 also induces a triangular fracture pattern, while larger friction values (0.2 and 0.3) generates square-like fracture patterns as shown in Fig. 4. However, when using larger mesh sizes, this distinction in fracture pattern is no longer observed. Nevertheless, the friction coefficient was set in 0.1 since it generates numerically the same collapse mode observed in experiments. At the same time, coefficient of 0.1 is reasonable for dynamic friction (Soares and Garbatov, 2015) and considering polished surfaces of both indenter and plate.

Fig. 4. Fracture patterns obtained from perforation test: a) experiment and b) FE modeling considering friction coefficients of 0.1 and 0.2.

3.3 Experiments on out-of-plane tearing loading

Tearing tests based on the third fracture mode were performed at different velocities. It consists in a crack opening test by separating two plate strips by applying forces in out-of-plane opposite directions as shown in Fig. 5 (Calle et al., 2017a). Similar to the tensile test, the specimen's strips are clamped to both testing machine grips. The vertical displacement of the upper grip will induce the crack opening/propagation along the specimen's body.

Fig. 5. Experiment setup and force responses obtained from tearing tests at different velocities.

For the tearing tests, two experimental setups were used: universal screw-driven and all-electric dynamic testing machines. Figure 5 presents the force-displacement responses for the tearing tests performed at several velocities. In a first stage, these strips are completely stretched in tension and a crack is generated. After that, the crack starts propagating showing a stable crack-opening force response. Some plate corrugation was detected besides the opened crack in all specimens. It is clearly observed that higher test velocities lead to higher tearing force-levels.

To evaluate the material failure strain at different tearing velocities a FE model was developed using uniform 0.5×0.5 mm shell elements and the coupled Power law plus Alves material model. An iterative process was performed to calibrate the plastic strains at failure in order to generate numerical equivalent force levels as the ones found in the experiments. This iterative procedure was repeated for each test velocity.

Mesh size influence on the failure strain was studied using different size shell elements. The crosshead velocity of 0.01 mm/s was used as reference for the test modeling so resulting in a nearly stable tearing force of 56.1

N. Different mesh sizes with equivalent length over thickness ratios (L_e/t) between 2 and 20 (0.5×0.5 , 1×1 , 1.667×2 , 2.5×3.125 , 5×3.571 and 5×5 mm) were considered for the analysis. Some of these models are presented in Appendix A.3.

Regardless of the mesh size, all FE models showed the same fracture pattern as obtained in the experiment, i.e. straight crack propagation along the strip. Given the sudden removal of the failing shell elements along the strip, high oscillations in the numerical force responses are generated. Larger shell element sizes lead to high force oscillations during crack propagation.

3.4 Stress triaxiality on failure

The evolution of the stress triaxiality for each one of the mechanical tests is presented in Fig. 6. Two ranges of strain rates are considered. Markers indicate the failing points by uniaxial tensile $\bigcirc$, perforation $\diamond$ and tearing Δ loadings. Small markers denote low strain rate and large markers high strain rate. Strain paths are built considering the first integration point to fail (Fig. 6). However, in the case of the tearing tests, both outer-shell integration points (1 & 5) are considered.

In the case of the uniaxial tensile tests in small specimens (UTT), the stress state was evaluated in the first element to fail, i.e. the central element in the middle failure section of the specimen. Initially, the stress triaxialities showed values close to $1/3$, then increasing up to values around 0.44. Regardless of the test velocity, all strain paths present the same behavior, barely differing in the failure strain value. Larger strain rates induce smaller failure strains.

For the perforation tests (PT), the stress state was also evaluated in the first element to fail, located in the contact area between the spherical indenter and plate surface. In general, the stress triaxiality presented steady values close to an equi-biaxial tension loading ($\eta = 2/3$) up to the rupture. Failure strains are significantly higher than in uniaxial tensile tests. As seen in uniaxial tensile tests, higher strain rates induce lower failure strains.

The stress state, observed in the tearing tests (TT), presented a different behavior when compared to the other tests. The analysis was performed in failing elements along the propagated crack at low and high strain rates. Unlike the other tests, strain path and failure occurrences vary according the integration points in the

1 same failing element. High stress triaxiality oscillations are observed in their strain paths. By contrast, larger
 2 strain rates induce larger failure strain, but in a smaller proportion.
 3 It is worth mentioning that the evaluation of stress states at failure is limited to the plane stress assumption
 4 of shell elements (so becoming unrealistic in case of triaxial loadings), and differs significantly from stress
 5 states evaluated in solid elements.

6
 7 Fig. 6. Failure strain versus triaxiality for all mechanical tests considering a mesh size of 0.5 mm ($L_e/t = 2$).

8 3.5 Influence of mesh size on failure

9 The influence of the mesh size on failure strain for each one of the three mechanical tests is shown in Fig. 7.
 10 This analysis comprises element sizes ranging from 0.5×0.5 mm to 5×5 mm (L_e/t ratios from 2 to 20
 11 respectively). Trend curves for the failure strain F_1 as a function of the L_e/t ratio are shown in Fig. 7 for each
 12 one of the mechanical tests. These curves do not have a particular form, they merely presented better data
 13 fitting. Normalized curves are also formulated as f_1 , considering a reference element length of $L_{e_0} = 0.5$ mm
 14 ($L_{e_0}/t = 2$), via

$$15 \quad f_1(L_e/t) = F_1(L_e/t)/F_1(L_{e_0}/t) \quad (13)$$

16 All tests showed strong sensitivity to mesh size. Uniaxial tensile test involves the uniform stretching of a
 17 material strip until the necking point and, from that point, the strain evolution is strongly dependent on the
 18 mesh size. In the perforation test, the plate failure is generated in circular areas of contact between the
 19 deformed plate surface and the rigid spherical indenter and these areas (and the induced strains) are sensitive
 20 to mesh size. In the tearing test, a crack is progressively opened and high plastic deformations and rupture of

the material occurs barely in a very small area near the crack tip. So, the induced plastic strains, close to the crack tip, are very sensitive to element size. When larger shell elements are used, the structural influence of the crack propagation in the whole shell area is diluted so yielding lower values of plastic strains at failure.

Fig. 7. Dependence of failure strain on mesh size considering a mesh size of $L_e = 0.5$ mm ($L_e/t = 2$).

3.6 Influence of strain rate on failure

Figure 8a shows the influence of the strain rate on failure strain obtained from the uniaxial tensile, perforation and tearing tests. Given the dynamic character of this analysis, only mechanical tests that induces strain rates above 0.01 s^{-1} in the failing elements are taken into account. The strain rate values are computed as the instantaneous strain rates (Eq. 5) obtained from the first element to fail in each one of the mechanical tests (as also regarded in the analysis of stress triaxiality in failure, Fig. 6).

The failure strains for each one of the mechanical tests were numerically calibrated at different strain rates by FE modeling considering a uniform shell element size of 0.5×0.5 mm ($L_e/t = 2$). Trend curves of the failure strain F_2 as a function of the natural logarithm of the strain rate are also presented in Fig. 8. Markers indicate the failing points calibrated by uniaxial tensile $\circ$, perforation $\diamond$ and tearing Δ tests. The size of the markers are such that the larger ones denote higher strain rates.

When the calibration of the strain at failure is based on uniaxial tensile or perforation tests, a decrease around 40% is observed in the plastic strain at rupture as the strain rate increases. In contrast to tensile and perforation tests, when the calibration of the strain at failure is based on the tearing tests, the plastic strain at failure increases up to 60% at high strain rates.

Figures 8b and 8c also present the F_2 curves of failure strain versus strain rate for element sizes of 2 and 5 mm ($L_{eq}/t = 8$ and 20 respectively) obtained by interpolation by weighting with f_1 . The interpolation curves agreed with the FE reproduction of perforation and tearing tests at different strain rates when using larger element sizes, 2 and 5 mm ($L_e/t = 8$ and 20 respectively).

Similarly to f_1 , normalized curves for F_2 are also formulated as f_2 considering a reference strain rate of $\dot{\epsilon}_0 = 0.2 \text{ s}^{-1}$ in the form

$$f_2(\dot{\epsilon}) = F_2(\dot{\epsilon})/F_2(\dot{\epsilon}_0) \quad (14)$$

Fig. 8. Strain rate sensitivity of failure strain considering a mesh size of $L_e = 0.5, 2$ and 5 mm (L_e/t ratios of 2, 8 and 20).

3.7 Material failure models

3.7.1 Plastic strain-based failure criterion

First, a simple plastic strain-based failure criterion (equivalent plastic strain at failure dependent on the mesh size and strain rate) is proposed for each one of the calibration mechanical tests (uniaxial tensile, perforation and tearing tests) as follows:

$$\varepsilon_f(L_e/t, \dot{\epsilon}) = f_1(L_e/t) f_2(\dot{\epsilon}) \varepsilon_{f_0} \quad (15)$$

where f_1 is the normalized failure strain as a function of the L_e/t ratio measured at a reference strain rate ($\dot{\epsilon}_0$), f_2 is the normalized failure strain as a function of the strain rate at a reference equivalent element length (L_{e_0}) and ε_{f_0} is the failure strain calibrated at a reference strain rate ($\dot{\epsilon}_0$) and a reference equivalent element length (L_{e_0}). An equivalent element length of $L_{e_0} = 0.5 \text{ mm}$, plate thickness $t = 0.25 \text{ mm}$ and a reference

1 strain rate around $\dot{\epsilon}_0 = 0.2 \text{ s}^{-1}$ were considered. Formulae $f_1(L_e/t)$, $f_2(\dot{\epsilon})$ and parameter $\epsilon_{f,0}$, for each one of
 2 the mechanical tests, are described in Appendix B. Figure 9 shows the failure locus for the plastic strain based
 3 on this failure criterion. The same failing points calibrated by uniaxial tensile, perforation and tearing tests,
 4 already presented in Fig. 6, are indicated by the markers $\circ$, $\diamond$ and Δ , respectively, in Fig. 9.

Fig. 9. Fracture locus for the plastic strain-based failure criterion in the triaxiality-failure strain space
 calibrated using uniaxial tensile tests (UTT), perforation tests (PT) and tearing tests (TT).

3.7.2 Stress state-based failure criterion

As observed previously in section 3.4, the stress states induced in the element just before its failure deletion are dependent on the type of mechanical test: uniaxial tensile, perforation or tearing.

The ductile fracture criterion proposed by Lou et al. (2012) was adopted in this work to include the triaxiality dependence on the material failure due to its simplicity and straightforward calibration. This criterion was constructed with consideration of damage accumulation induced by nucleation, growth and shear coalescence of voids (as stated by Lou et al., 2012). Despite its concept be grounded on solid elements, it was implemented for plane stress applications, i.e., shell elements. The failure criterion is given as follows:

$$\left(\frac{2\tau_{max}}{\sigma_{eq}}\right)^{C_1} \left(\frac{(1+3\eta)}{2}\right)^{C_2} \epsilon_f = C_3 \quad (16)$$

or

$$\epsilon_f = C_3 \left(\frac{2\tau_{max}}{\sigma_{eq}}\right)^{-C_1} \left(\frac{(1+3\eta)}{2}\right)^{-C_2} \quad (17)$$

$$\tau_{max} = \frac{1}{2} \max\{|\sigma_1 - \sigma_2|, |\sigma_2 - \sigma_3|, |\sigma_3 - \sigma_1|\} \quad (18)$$

1 where $\langle x \rangle = x$ if $x > 0$ or $\langle x \rangle = 0$ if $x \leq 0$, τ_{max} is the maximum shear stress, σ_{eq} is the Von Mises stress,
 2 η is the stress triaxiality, ε_f is the failure strain and C_1 , C_2 and C_3 are material parameters.

3 In order to include the mesh size and strain rate sensitivity, this criterion was coupled with average normalized
 4 trend curves $\bar{f}_1$ and $\bar{f}_2$ by weighting the parameter C_3 as follows

$$5 \quad \varepsilon_f = \bar{f}_1(L_e/t) \bar{f}_2(\dot{\varepsilon}) C_3 \left(\frac{2 \tau_{max}}{\sigma_{eq}} \right)^{-C_1} \left(\frac{\langle 1+3\eta \rangle}{2} \right)^{-C_2} \quad (19)$$

6 The stress triaxialities – induced by the uniaxial tensile, perforation and tearing tests – cover uniformly the
 7 whole triaxiality range observed in failure occurrence of thin shell elements, i.e. from pure shear, uniaxial
 8 tension, plain strain to equi-biaxial tension, as shown in Fig. 6. So, the average normalized trend curves can be
 9 obtained by averaging the curves obtained from the three mechanical tests as follows:

$$10 \quad \bar{f}_1(L_e/t) = \frac{1}{3} \{ f_1^{UTT} + f_1^{PT} + f_1^{TT} \} (L_e/t) \quad (20)$$

$$11 \quad \bar{f}_2(\dot{\varepsilon}) = \frac{1}{3} \{ f_2^{UTT} + f_2^{PT} + f_2^{TT} \} (\dot{\varepsilon}) \quad (21)$$

12 where UTT, PT and TT denote trend curves f_1 and f_2 obtained from uniaxial tensile tests, perforation tests and
 13 tearing tests respectively.

14 The material parameters $\{C_1, C_2, C_3\}$ are $\{3.5, -1.2, 0.71\}$. They were calibrated based on failure strains
 15 obtained from uniaxial tensile, perforation and tearing tests and considering an equivalent element length of
 16 $L_{e_0} = 0.5$ mm, plate thickness $t = 0.25$ mm ($L_{e_0}/t = 2$) and a reference strain rate around $\dot{\varepsilon}_0 = 0.2$ s⁻¹.

17 As a result, it can be observed in Fig. 10 that fitting the curve through all the points was not possible. While a
 18 good adjustment was obtained for perforation tests at low and high strain rates (0.2 and 900 s⁻¹ respectively),
 19 a good match was only achieved for uniaxial tensile test at low strain rate. Besides, it is very difficult to achieve
 20 a good correspondence with tearing tests points given their reverse behavior: higher strain rates lead to higher
 21 failure strains. However, a lower ceiling was considered for shear failure ($0 < \eta < 1/3$) irrespective of being at
 22 low or high strain rates.

23 This failure criterion was implemented in the commercial code Abaqus/CAE 2016 as “Ductile Damage” model
 24 by listing the fracture strain as a function of stress triaxiality and strain rate. No damage softening was
 25 considered by setting the parameter “displacement at failure” to zero.

1 The failure locus for the stress state based on this failure criterion is shown in Fig. 10. The same failing points
2 calibrated by uniaxial tensile, perforation and tearing tests, already presented in Fig. 6, are indicated by the
3 markers $\circ$, $\diamond$ and Δ respectively in Fig. 10.

4
5 Fig. 10. Fracture locus for the stress state-based failure criterion (SSBC) in the triaxiality-failure strain space.

6 4. Case study: Miniature ship grounding test

7 4.1 Experiment

8 A ship grounding accident was reproduced experimentally in a 1:100 reduced scale. A ship grounding scenario
9 involves a combined vertical/horizontal movement of the ship bottom structure over a rigid sharp rock in the
10 seabed.

11 A ship bottom structure of an oil tanker was manufactured in 0.01 scale reduction using the 0.25 mm thick
12 mild steel sheets tested in the previous section. The ship bottom structure consists in a double plate panel
13 structure of $210 \times 230 \times 23$ mm with four horizontal and four vertical web frames uniformly distributed.
14 Secondary stiffening structures and cutouts were disregarded. All parts were laser cut, mounted and laser
15 welded.

16 The experimental arrangement consists in a ship bottom structure mounted in a horizontal slider colliding at
17 2.47 m/s against a rigid sharp indenter at an angle of 19.8° (so inducing a relative vertical/horizontal
18 penetration). The rest of the ship body was simplified as a mass in the mobile platform guided in the horizontal
19 slider. This platform, mass and the ship bottom structure resulted in a total mass of 51.04 kg (Calle et al.,
20 2017a).

The experimental measurement of the structure motion during the test time (about 0.1 s) was performed using a high-speed digital camera recording at 20,000 frames per second and a motion tracking software. From the high-speed camera images, the structure position, $x(t)$, is measured directly from the structure frame position, which is assumed to move as a rigid body. Next, the velocity and acceleration of the structure are obtained by the first and second time derivative of the position signal respectively, such as $\dot{x} = dx(t)/dt$ and $\ddot{x} = d^2x(t)/dt^2$. The horizontal force is finally evaluated by multiplying acceleration by the total platform mass (M) in the form: $F(t) = M \ddot{x}(t)$.

Fig. 11. Scheme and experimental setup of ship grounding test in miniature.

As a result, the outer plate of the ship bottom structure was torn locally by the frontal face of the rigid obstacle so causing a predominant global tearing collapse of the structure as shown in Fig. 12. The plate cut off by the obstacle was folded and pressed against the last transversal plate. The area of the torn plate increased progressively after the second transversal web frame, Fig. 12, but it was continually delimited by the horizontal web frames. The internal transversal web frames showed a structural collapse partially stretched before being torn by the rigid obstacle, Fig. 13.

Occurrence of rupture in welded joints was not significant in the structural collapse as a whole, barely the joint between the transversal reinforcement and the inner plate showed complete weld fillet detachment, Fig. 12.

Fig. 12. Collapsed ship bottom structure: overall collapse with obstacle and torn outer plate.

The resulting reaction force (in the horizontal direction) from the ship grounding experiment is directly related with the inner collapse of the double plate bottom structure as shown in Fig. 13. The initial contact between the rigid obstacle and the miniature ship bottom structure occurs in point ① at zero force, the outer plate is stretched up to a maximum peak force ② just before outer plate perforation (Similarly, a small peak force is detected when perforates the inner plate in ⑤). The reaction force remains close to steady during the outer plate tearing process: ② to ⑤. Small force peaks are observed when the rigid obstacle contacts each one of the structural joints in outer plate, ③ and ④, and large force peaks in both outer and inner plates, ⑥ and ⑦. Finally, the maximum displacement of the structure is achieved in ⑧ when all the kinetic energy is consumed and the reaction force falls to zero.

Fig. 13. Experimental reaction force and correlation with internal structure collapse.

4.2 FE modeling

A finite element model of the ship bottom structure was created in the commercial code Abaqus/CAE 2016 considering the mid-plane dimensions of the structure and 2×2 mm shell elements for both mild steel and welded joints as shown in Fig. 14. The indenter and the mobile base for the ship bottom structure were modeled as rigid bodies and a general friction coefficient of 0.1 was considered between all contact surfaces.

Fig. 14. Mesh of ship bottom structure: outer plate and inner plate views.

1 The elasto-plastic behavior of the base material (0.25 mm thick mild steel sheet) was modeled using the
2 coupled Power law plus Alves model seen previously in section 2. The laser-welded joints were modeled using
3 the same elasto-plastic behavior of the base material, but considering a failure strain of 0.2064 calibrated
4 using tensile test experiments in welded strip samples and considering plastic work equivalence (similar to the
5 procedure for failure in tensile test calibration, section 3.1).

6 The performance of the plastic strain-based failure criteria with strain rate and mesh size sensitivity (shown in
7 Fig. 9) for the base material is evaluated. These criteria calibrated from tensile, perforation and tearing tests
8 are named as UTT, PT and TT criteria respectively. The experimental response is here called as EXP.

9 Figure 15 also presents the force-displacement responses generated numerically by employing the UTT, PT
10 and TT failure criteria with strain rate sensitivity shown in Fig. 9. However, none of those generated compatible
11 results with that obtained in the experiment. Among these criteria, TT failure criterion produced force levels
12 closer to that obtained experimentally, probably because the plate tearing (crack propagation) is dominant in
13 the overall structure collapse as seen in Fig. 15.

14 The performance of the stress state-based failure criterion (SSBC) coupled with strain rate and mesh size
15 sensitivities, Eq. (19), showed to be superior to that obtained by all the plastic strain-based failure criteria,
16 regardless of the type of mechanical test.

17 When analyzing the kinematic response, a slight difference between the experimental and numerical velocity
18 histories is observed from 100 to 110 mm displacement as pointed with arrows in Fig. 16. This divergence was
19 mainly generated by the inaccurate modeling of the accumulated folded material that, in turn, generated a
20 slight crack propagation deviation in the ship bottom structure. Consequently, the modeled crack path
21 demanded less energy for penetration so inducing an increase in the final obstacle penetration of about +4.7%.

Fig. 15. Force and velocity responses obtained numerically using UTT, PT and TT failure criteria.

Fig. 16. Force and velocity responses obtained numerically using the stress state-based criterion (SSBC).

Fig. 17. Sectioned views of collapsed ship bottom structure: right and left sections.

4.3 Strain rate

For marine structures subjected to collision and grounding, certain areas of their structural members can undergo plastic deformation at high strain rates, but not necessarily all these areas can lead to material fracture. For instance, the local buckling of a structural member can generate large localized plastic deformation in the hinge area but not always leading to material rupture.

However, it is very common to observe material rupture in intersection joints of marine structures subjected to collision because of the high stress concentration. Storheim and Amdahl (2017) evaluated numerically the crushing of a bulb tip, so resulting in a strain rates associated with plastic strain between 5 and 15 s⁻¹. They also stated that strain rates can achieve values up to 100 s⁻¹ in ship collision simulations considering shell elements with L_e/t ratios between 5 and 10. However, they did not specify if the elements that undergo high strain rates achieve rupture.

Furthermore, Ko et al. (2018) performed FE modeling of the lateral ship collision with varying collision speeds (from 0.5 to 20 knots) so resulting in the rupture of the side shell of the struck ship at strain rates from 0.3 to 30 s⁻¹. L_e/t ratios around 5 were considered in the fine mesh of structural members near the collision location. Here, in the numerical modeling of the miniature ship grounding test with uniform shell elements with a L_e/t ratio of 8, strain rates in the range of 110-340 s⁻¹ are observed in failing elements subjected mainly to tearing and stretching. When bringing this miniature experiment to real scale using similarity laws (scaling formulae detailed by Oshiro and Alves, 2009; Oshiro et al., 2017), this strain rate range corresponds to 1.1-3.4 s⁻¹. This range is compatible with that obtained from FE modeling of the grounding experiment but in real scale, which generates strain rates in failing elements in the range of 2.4 to 4.9 s⁻¹.

All these strain rate ranges resulting from ship collision events are compared in Fig. 18. The strain rate range obtained from the miniature ship grounding test showed to be compatible with that obtained from real-scale ship collision studies.

The high levels of strain rates induced in the miniature ship grounding experiment, when compared against the strain rate levels induced in real scale events, ended up intensifying the influence of the strain rate

sensitivity on the failure criterion and, consequently, on the structural response. So, as lower strain rates are expected in real scale tests, lower influence of the strain rate sensitivity is also expected.

The strain rate ranges comprised by the tensile, perforation and tearing tests at different test velocities are also included in Fig. 18. The strain rates obtained by the mechanical tests cover, almost completely, the strain rate ranges observed in miniature and real scale ship collision/grounding events.

Fig. 18. Strain rates ranges obtained from mechanical tests and FE simulations of ship collision/grounding.

5. Conclusions

In this work the effect of the strain rate on the failure strain and the type of mechanical test used for the failure calibration are evaluated experimentally and numerically. Then, a miniature ship grounding experiment was FE modeled to corroborate the real influence of these aspects when modeling collision events. A double-plate bottom structure of a tanker ship, in 1:100 reduction scale, was considered for the experiment. This structure was made of a 0.25 mm thickness mild steel plate, cut and welded by laser.

The effect of the strain rate on the material fracture of mild steel plates when subjected to different mechanical tests was evaluated by uniaxial tensile, perforation and tearing experiments at different loading rates. The aim of these singular mechanical tests is to cover, to the extent possible, the fracture modes observed in marine structures subjected to collision/grounding related to crack initiation and propagation.

In general, experiments related to crack initiation (uniaxial tensile and perforation tests) showed failure strains substantially higher than that related to crack propagation (tearing tests). Uniaxial tensile and perforation tests presented a decrease in the failure strains at high strain rates while tearing tests presented an opposite effect, i.e., failure strains' increase together with strain rates. Yet all tests presented a decrease in the failure strains with the increase of the element mesh size.

Different to that observed in uniaxial tensile and perforation tests ($0.33 < \eta < 0.66$), tearing tests induced strong oscillations in the stress triaxiality in failing elements before fracture occurrence. In spite of these oscillations, it can be stated that the stress state induced in tearing tests is located more heavily between pure shear and uniaxial tensile regimes ($0.0 < \eta < 0.33$).

The collapse mode observed in the miniature ship bottom structure, generated in the ship grounding experiment, is an alternating combination of stretched and torn structural members along the penetration path, so exhibiting clearly material rupture governed by both crack initiation and propagation.

Two failure criteria were employed to FE model the miniature ship grounding experiment: a plastic strain-based criterion and a stress state-based criterion, both involving sensitivity to element size and strain rate.

The plastic strain-based failure criterion failed in reproducing accurately the structural response and collapsed mode obtained from the miniature ship grounding experiment. When the failure criterion is calibrated based on mechanical tests related to crack initiation (uniaxial tensile or perforation tests), it tends to overestimate the demanded energy for material fracture, while, when calibration is based on crack propagation (tearing tests), it tends to underestimate this energy.

In addition, the stress state-based failure criterion successfully managed to include all these effects through stress triaxiality dependence so resulting in a more accurate structural response and collapse mode. To extend effectively this criterion to shipbuilding steels, new tests would be required focusing specifically these materials.

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7. Data availability

The raw experimental data from all tensile, perforation and tearing tests presented in this work are available to download from <http://dx.doi.org/10.17632/8k43xh5fv8.2>.

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Appendixes

Appendix A. Mesh configurations for tensile, perforation and tearing tests using different mesh sizes

A1. Mesh models for tensile test with $L_e/t = 4, 6.67, 10, 16.33$ and 20 .

A2. Mesh models for perforation tests with $L_e/t = 2, 8$ and 20 .

A3. Mesh models for tearing tests with $L_e/t = 4, 7.3$ and 11.18 .

Appendix B. Trend curves for equivalent plastic strain at failure for tensile, perforation and tearing tests dependent on mesh size and strain rate

Tensile tests	$F_1^{UTT}(L_e/t) = -0.00924 \cdot L_e/t + 0.6175$ <p>obtained at quasi-static conditions (crosshead velocity of $v_0 = 0.0025$ mm/s)</p> $F_1^{UTT}(L_{e_0}/t = 2.0) = 0.5990$ $f_1^{UTT}(L_e/t) = -0.015426 \cdot L_e/t + 1.03087$ $F_2^{UTT}(\dot{\epsilon}) = -0.02044942 \cdot \ln(\dot{\epsilon}) + 0.51242252$ <p>obtained at a reference element length of $L_{e_0} = 0.5$ ($L_{e_0}/t = 2.0$)</p> $F_2^{UTT}(\dot{\epsilon}_0 = 0.2 \text{ s}^{-1}) = 0.545335$ $f_2^{UTT}(\dot{\epsilon}) = -0.0375 \cdot \ln(\dot{\epsilon}) + 0.93965$ $\epsilon_{f0}^{UTT} = 0.545335$
Perforation tests	$F_1^{PT}(L_e/t) = 0.962022 \cdot e^{-0.06265 \cdot L_e/t}$ <p>obtained at quasi-static conditions (indenter velocity of $v_0 = 0.0025$ mm/s)</p> $F_1^{PT}(L_{e_0}/t = 2.0) = 0.8487$ $f_1^{PT} = 1.13348 \cdot e^{-0.06265 \cdot L_e/t}$ $F_2^{PT}(\dot{\epsilon}) = -0.03121 \cdot \ln(\dot{\epsilon}) + 0.88224$ <p>obtained at a reference element length of $L_{e_0} = 0.5$ ($L_{e_0}/t = 2.0$)</p> $F_2^{PT}(\dot{\epsilon}_0 = 0.2 \text{ s}^{-1}) = 0.93247$ $f_2^{PT}(\dot{\epsilon}) = -0.03347 \cdot \ln(\dot{\epsilon}) + 0.94613$ $\epsilon_{f0}^{PT} = 0.93247$

Tearing tests	$F_1^{TT}(L_e/t) = 0.44947317(L_e/t)^{-0.67538904}$ obtained at quasi-static conditions (crosshead velocity of $v_0 = 0.01$ mm/s) $F_1^{TT}(L_{e_0}/t = 2.0) = 0.5764$ $f_1^{TT}(L_e/t) = 0.7798 (L_e/t)^{-0.67538904}$ $F_2^{TT}(\dot{\epsilon}) = 0.01151233 \cdot \ln(\dot{\epsilon}) + 0.30599155$ obtained at a reference element length of $L_{e_0} = 0.5$ ($L_{e_0}/t = 2.0$) $F_2^{TT}(\dot{\epsilon}_0 = 0.2 \text{ s}^{-1}) = 0.28746$ $f_2^{TT}(\dot{\epsilon}) = 0.04005 \cdot \ln(\dot{\epsilon}) + 1.064455$ $\varepsilon_{f0}^{TT} = 0.28746$
Average trend curves	$\bar{f}_1(L_e/t) = -0.12120804 \cdot \ln(L_e/t) + 0.66460069$ $\bar{f}_2(\dot{\epsilon}) = -0.01338240 \cdot \ln(\dot{\epsilon}) + 0.56688443$