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Anisotropic thermoelectric properties associated with dimensional crossover in quasi-one-dimensional SrNbO$_3$,$d+4$ ($d \sim 0.03$)

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We have grown large single crystals of SrNbO$_3$,$d+4$ ($d \sim 0.03$) with $n = 5$ in the homologous series Sr$_n$NbO$_{3n+2}$ by using a traveling solvent floating-zone method and measured resistivity, thermopower, and thermal conductivity along all crystallographic axes. The thermoelectric properties are found to be highly anisotropic, which reflects a quasi-one-dimensional electronic structure. In particular, the thermopower along the $b$ axis is $-170 \mu$V/K at 300 K, which is 1 order of magnitude higher than the $-15 \mu$V/K along the $a$ axis and $-25 \mu$V/K along the $c$ axis. A possible origin of the high anisotropy in the thermopower is discussed in terms of dimensional crossover associated with a structural modification.

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I. INTRODUCTION

Thermoelectric oxides are promising materials for recovering electric energy from waste heat. Recent developments in nanotechnology enable the achievement of a thermoelectric performance higher than that of conventional thermoelectrics. Since the enhancement of the dimensionless figure of merit $ZT \equiv \frac{S^2 T}{\rho \kappa}$ ($S$, thermopower; $\rho$, resistivity; $\kappa$: thermal conductivity; and $T$, temperature) in nanostructures was predicted by Hicks and Dresselhaus in 1993, artificial structures with low dimensionality have been extensively studied. A nanostructure such as a superlattice or nanowire can show good thermoelectric properties owing to the reduced thermal conductivity by phonon scattering at the artificially controlled boundary and the enhanced thermopower by a steep change in the density of states at the Fermi level. Indeed, the Bi$_2$Te$_3$/Sb$_2$Te$_3$ superstructure exhibits high performance above $ZT = 2.4$, and also recent studies on Si nanowires show a performance 50 times higher than that of bulk Si. The two-dimensional (2D) electron gas induced at the interface between SrTiO$_3$ and TiO$_2$ also displays a high thermopower, of the order of 1 mV/K.

Not only artificial nanostructures but also bulk oxides with a low-dimensional electronic structure have been known to be promising thermoelectric materials since 1997. Layered cobalt oxide Na$_x$CoO$_2$ and misfit layered cobalt oxides exhibit a $ZT$ of the order of unity at high temperatures. Their large thermopowers are successfully explained by both band calculation and the extended Heikes formula. Compared with the extensive study of layered oxides, there has been little research on the thermoelectricity of quasi-one-dimensional (Q1D) oxides, because they tend to be insulating even at room temperature and sometimes exhibit a charge density wave (CDW) or spin density wave (SDW) due to the nesting of the sheet-like Fermi surfaces, which is a disadvantage for thermoelectricity.

We have studied thermoelectric properties of Q1D Hollandites Ba$_{12}$Rh$_8$O$_{16}$ and K$_2$Ru$_8$O$_{16}$ and found that Ba$_{12}$Rh$_8$O$_{16}$ exhibits the large power factor ($P \equiv \frac{S^2}{\rho}$) of 30 $\mu$W/cm K$^2$ at 75 K, comparable to that seen in Na$_x$CoO$_2$ at 300 K. However, the magnitude of the thermopower is about 20 $\mu$V/K at 300 K for both Hollandites, and enhancement of the thermopower due to the Q1D electronic structure was not observed. The low anisotropy in the resistivity ($\rho_{\perp}/\rho_{\parallel} \sim 3$ at 300 K) in these systems implies a three-dimensional (3D)-like electronic structure rather than a Q1D one. Thus, a Q1D material with a high anisotropy in conduction is desirable for the enhanced thermopower.

SrNbO$_3$,$d+4$ with Nb$^{4+}$ ($4d^0$) is a good Q1D conductor. SrNbO$_3$ belongs to a homologous series, Sr$_n$NbO$_{3n+2}$ ($n = 5$; Sr$_5$Nb$_5$O$_{17}$), and is derived from the 3D network.
grown single crystals of SrNbO$_3$ were prepared by a solid-state reaction. Stoichiometric amounts of SrCO$_3$ and Nb$_2$O$_5$ were mixed, and the mixture was sintered at 1300 °C for 12 h. Then the product was finely ground, pressed into a bar shape with a 5-mm diameter and 5-cm length, and sintered at 1300 °C for 12 h. Large single crystals with a typical dimension of 5 × 5 × 1 mm$^3$ were successfully grown.

The x-ray diffraction pattern of the single crystal along the c axis was checked using a standard diffractometer with CuK$_\alpha$ radiation as an x-ray source in the θ-2θ scan mode (Rigaku MiniFlexII). Synchrotron powder x-ray diffraction measurements at a wavelength of 0.7749 Å were carried out at the BL-8A at the Photon Factory (KEK), controlling the temperature from 80 to 300 K, and structural refinements were performed by Rietveld analysis (RIETAN-FP). The oxygen stoichiometry was determined by means of thermogravimetry (TG) described elsewhere. The orientations of the crystals were determined by backreflection Laue x-ray diffraction.

Resistivity was measured by a four-probe method in a liquid He cryostat. Gold wires were carefully attached to the sample using silver paste (DuPont 4922) as electrodes. Thermopower was measured using a steady-state technique in a liquid He cryostat with a copper-constantan thermocouple to detect a small temperature gradient of about 1 K/cm. Thermal conductivity was measured by a steady-state technique with a chromel-constantan differential thermocouple to detect a temperature gradient. One edge of the crystal was glued to a cold head of a closed cycle refrigerator with varnish (GE 7031), while a resistive heater (~120 Ω; KYOWA strain gauges) was attached on the opposite edge. The error in the heat current estimation should be below 0.15% using a chromel wire of 50-μm diameter and 2-cm length. To avoid leakage of the heat current by radiation and conduction through the air, we attached a Cu radiation shield and evacuated the air down to 10$^{-4}$ Pa. Heat loss was evaluated to be ~16% along the c axis, which is greater than the ~8% along the a axis and ~5% along the b axis at 300 K.

## III. RESULTS AND DISCUSSION

### A. Sample characterization and transport properties

Figure 2(a) shows the x-ray diffraction pattern of a SrNbO$_3$ single crystal along the c axis. (00l) peaks are clearly observed. The c-axis length was evaluated to be 32.47 Å, which is very close to that reported in the preceding work. As shown in Fig. 2(b), the synchrotron x-ray diffraction pattern of a ground single crystal was well fitted by orthorhombic symmetry (Pnma; Int. Tables A-58-1) with lattice parameters of $a = 3.988$ Å, $b = 5.676$ Å, and $c = 32.467$ Å. From TG measurement, the oxygen content which is 1 order of magnitude higher than those along the other axes. A possible origin of the enhanced thermopower along the b axis is discussed in terms of dimensional crossover associated with a structural modification.

### II. EXPERIMENTS

High-quality single crystals of SrNbO$_3$ were grown by a traveling solvent floating zone method with a growth velocity of 10 mm/h in a mixed-gas flow of Ar (97%) and H$_2$ (3%) at 0.2 L/min. Feed rods of fully oxidized SrNbO$_3$ were prepared by solid-state reaction. Stoichiometric amounts of SrCO$_3$ and Nb$_2$O$_5$ were mixed, and the mixture was sintered at 1300 °C for 12 h. Then the product was finely ground, pressed into a bar shape with a 5-mm diameter and 5-cm length, and sintered at 1300 °C for 12 h. Large single crystals with a typical dimension of 5 × 5 × 1 mm$^3$ were successfully grown.
ANISOTROPIC THERMOELECTRIC PROPERTIES

**FIG. 2.** (Color online) (a) X-ray diffraction pattern along the c axis and (b) synchrotron powder x-ray diffraction pattern of a ground single crystal.

$d$ was evaluated to be 0.03. Accordingly, we use SrNbO$_{3.43}$ as the chemical formula hereafter.

Figure 3(a) shows the temperature dependence of the resistivity along the crystallographic axes ($\rho_a$, $\rho_b$, and $\rho_c$). The magnitude of the resistivity is 7, 110, and 840 m$\Omega$cm at 300 K for the $a$, $b$, and $c$ axes, respectively, showing an anisotropy of $\rho_b/\rho_a \sim 15$ in the $ab$ plane, and $\rho_c/\rho_a \sim 120$ in the $ac$ plane. This anisotropic behavior is the same as seen in SrNbO$_{3.41}$. On the other hand, the magnitudes of the resistivities are several times larger than those of SrNbO$_{3.41}$. This difference is caused by the different carrier concentrations, namely, different valences of Nb ions, Nb$^{4.82+}$($4d^{0.18}$) for SrNbO$_{3.41}$ and Nb$^{4.86+}$($4d^{0.14}$), for SrNbO$_{3.43}$. Indeed, SrNbO$_{3.45}$, with the higher valence of Nb$^{4.8+}$($4d^{0.1}$), exhibits resistivities higher than those of SrNbO$_{3.43}$. As shown in the inset in Fig. 3(a), the temperature dependence of the resistivity is described over a wide range of temperatures by the activation transport as $\rho = \rho_0 \exp(E_g/k_B T)$, where $\rho_0$, $E_g$, and $k_B$ represent a constant, the activation energy, and the Boltzmann constant. At around 300 K, $E_g$ was evaluated to be 29.7, 26.4, and 19.9 meV for the $a$, $b$, and $c$ axes, respectively. With decreasing temperature, $\rho(T)$ changes at around $T^{*} = 100$ K, where $\rho(T)$ displays the change in the activation energy. In particular, $\rho_b$ increases significantly above $T^{*}$. According to the Mott formula, $S$ is described as $S = \pi^2 k_B^2 T (\ln e / e)^2$, where $e$ and $\sigma$ represent the electron charge and electric conductivity, respectively. Here, the conductivity $\sigma$ is related to the electronic state at the Fermi level. Thus, the change in $\rho(T)$ and $S(T)$ at $T^{*}$ implies a modification of the electronic state of SrNbO$_{3.43}$. Similar behaviors are also observed in the layered-cobalt oxysulfide Sr$_2$Cu$_2$CoO$_2$S$_2$, in which electronic inhomogeneity is proposed at about the temperature where the thermopower increases, and in the Q1D oxide PrBa$_2$Cu$_4$O$_8$, in which dimensional crossover is suggested. Note that the thermopower along the $c$ axis differs in magnitude from that along the $a$ and $b$ axes.

Figure 3(b) shows the temperature dependence of the thermopower along the $a$, $b$, and $c$ axes ($S_a$, $S_b$, and $S_c$), respectively. The thermopower is negative along all the directions, and the magnitude (of the thermopower) along the $b$ axis is found to be 170 $\mu$V/K at 300 K, which is about 10 times higher than the 15 $\mu$V/K for the $a$ axis and 20 $\mu$V/K for the $c$ axis. Though 1D conductors generally show such an anisotropic thermopower due to the anisotropic electronic structure, we would like to emphasize that the coexistence of a high thermopower that is typical for a semiconductor and a low thermopower for a conventional metal is a significant characteristic of this system. The temperature dependence of the thermopower seems to change at around $T^{*} = 100$ K, where $S_b(T)$ displays the change in the activation energy. In particular, $S_b$ increases significantly above $T^{*}$. According to the Mott formula, $S$ is described as $S = \pi^2 k_B^2 T (\ln e / e)^2$, where $e$ and $\sigma$ represent the electron charge and electric conductivity, respectively. Here, the conductivity $\sigma$ is related to the electronic state at the Fermi level. Thus, the change in $\rho(T)$ and $S(T)$ at $T^{*}$ implies a modification of the electronic state of SrNbO$_{3.43}$. Similar behaviors are also observed in the layered-cobalt oxysulfide Sr$_2$Cu$_2$CoO$_2$S$_2$, in which electronic inhomogeneity is proposed at about the temperature where the thermopower increases, and in the Q1D oxide PrBa$_2$Cu$_4$O$_8$, in which dimensional crossover is suggested. Note that the thermopower along the $c$ axis differs in magnitude from that reported in Ref. 21. The difference could originate from the different density of states along the $c$ axis.

Figures 4(a)–4(c) show the temperature dependence of the thermal conductivity along the $a$, $b$, and $c$ axes ($\kappa_a$, $\kappa_b$, and $\kappa_c$), respectively. As seen for $\rho$ and $S$, $\kappa$ also exhibits a high anisotropy; $\kappa_a$ is 6.0 W/mK at 300 K, which is 3 times larger than the $\kappa_b$ of 2.1 W/mK and 10 times larger than...
the \( \kappa_c \) of 0.6 W/mK. The total thermal conductivity \( \kappa_{\text{tot}} \) is the sum of the lattice thermal conductivity \( \kappa_{\text{ph}} \) and the electron thermal conductivity \( \kappa_{\text{el}} \) \( \left( \kappa_{\text{tot}} = \kappa_{\text{ph}} + \kappa_{\text{el}} \right) \). In this system, \( \kappa_{\text{el}} \) is negligibly small because of the rather low electric conductivity. As a result, this anisotropy comes from \( \kappa_{\text{ph}} \). Such a high anisotropy in \( \kappa_{\text{ph}} \) is also seen in the 2D superlattice or Q1D organic compound, where the mean free paths of phonons are different along crystallographic axes.\(^{30,31} \) As shown in Fig. 1(a), phonons may easily propagate along the \( a \) axis due to straight Sr-Sr-Sr and Nb-O-Nb-O networks, while a zigzag network along the \( b \) axis and a layered structure along the \( c \) axis can be effective scatterers. The possibility that there is also substantial anisotropy in the phonon dispersion (anisotropy in sound velocity) is an open question. In addition to the anisotropy, we would like to emphasize that the phonon mean free path is very weakly temperature dependent. The thermal conductivity data show no peak structure at low temperatures, which implies that the phonon-phonon scattering rate is high. The high scattering rate can be attributed to defects in the crystal. Table I lists \( S, \rho, \kappa, \) and dimensionless figure of merit \( ZT \) at 300 K along all crystallographic directions. Owing to the high thermopower along the \( a \) axis, \( ZT \) reaches \( 3.7 \times 10^{-3} \) along the \( b \) axis.

**B. Structure-related thermoelectric properties**

As shown in Fig. 3, SrNbO\(_3\) undergoes a metallic-like state below 150 K: (1) the activation energy decreases with decreasing temperature; and (2) \( \frac{dS}{dT} \) decreases with decreasing temperature, in particular, along the \( b \) axis. These results indicate that SrNbO\(_3\) undergoes a metallic-like state at low temperatures compared with the weak semiconducting state above 150 K. This situation is rather strange, because Q1D material often displays a CDW or SDW state associated with insulating behavior, which cannot explain our observations. Recently, Campos et al. revealed that the small gap in resistivity at low temperatures is not attributable to CDW, showing heat capacity and thermal-expansion-coefficient measurements, and reported that power-law fitting is reasonable, indicating a possible signature of Luttinger liquid behavior.\(^{23} \)

To elucidate the possible origin of the crossover, the crystal structure was analyzed by Rietveld refinement using synchrotron powder x-ray diffraction data between 70 and 300 K. Figure 5 shows the temperature dependences of the lattice parameters. A structural phase transition was not observed in the measured temperature range, though the temperature dependence of the lattice parameters changes slightly below 150 K. This behavior is consistent with the temperature dependence of the linear thermal expansion of SrNbO\(_3\).\(^{23} \)

Now we mainly focus on the transport properties along the \( b \) axis where the thermopower rapidly increases at around \( T^* \). Figure 6(a) depicts the projection along the \( a \) axis of SrNbO\(_3\). There are three crystallographic Nb sites \([\text{Nb}(1), \text{Nb}(2), \text{and Nb}(3)]\) in SrNbO\(_3\). According to the band calculations,\(^{32,33} \) the predominant contribution to the density of states at \( E_F \) is attributed to the Nb(1) site situated in the middle of the slab [see Fig. 6(a)], while the contributions of the other two Nb sites are considerably small. The Nb(1)O\(_6\) octahedra are the least distorted ones, with the Nb atoms situated in the center of the octahedra, while the Nb(2) and Nb(3) atoms are considerably displaced away from the central position of the NbO\(_6\) octahedron. Consequently, the Q1D transport is related to the central chain with the least distorted octahedra in the slab. Thus, the temperature dependences of the bond lengths of Nb-O and/or Nb-O-Nb bond angles of Nb(1)O\(_6\) and Nb(3)O\(_6\) octahedra are important for understanding the transport properties along the \( b \) axis.

Figures 6(b)–6(e) show the temperature dependence of Nb(1)-O(\( x \)) \((x = 1, 2)\) and Nb(3)-O(\( x \)) \((x = 1, 2)\) distances and Nb(1)-O(1)-Nb(3) and Nb(1)-O(2)-Nb(3) angles, respectively. Both angles decrease monotonically, which signifies rotations of NbO\(_6\) octahedra with decreasing temperature. On the other hand, Nb(1)-O(2) and Nb(3)-O(2) bond lengths change significantly at around 150 K. Now let us focus on Fig. 6(b),

---

**TABLE I.** \( S \) (\( \mu \text{V/K} \)), \( \rho \) (m\( \Omega \text{cm} \)), \( \kappa \) (W/mK), and \( ZT \) values at 300 K.

<table>
<thead>
<tr>
<th>Axis</th>
<th>( S )</th>
<th>( \rho )</th>
<th>( \kappa )</th>
<th>( ZT )</th>
</tr>
</thead>
<tbody>
<tr>
<td>( a )</td>
<td>15</td>
<td>7</td>
<td>6.0</td>
<td>1.6 \times 10^{-4}</td>
</tr>
<tr>
<td>( b )</td>
<td>170</td>
<td>110</td>
<td>2.1</td>
<td>3.7 \times 10^{-3}</td>
</tr>
<tr>
<td>( c )</td>
<td>20</td>
<td>840</td>
<td>0.6</td>
<td>2.4 \times 10^{-5}</td>
</tr>
</tbody>
</table>
The Nb(1)-O(1) bond length is equal to the Nb(1)-O(2) bond length, which shows that Nb(1)O₆ is a perfect octahedron. On the other hand, the Nb(1)-O(1) bond length is significantly larger than the Nb(1)-O(2) bond length at 80 K, reflecting a distortion of the Nb(1)O₆ octahedron [see Fig. 6(a)]. Correspondingly, the difference between the Nb(3)-O(1) and the Nb(3)-O(2) bond lengths becomes smaller with decreasing temperature, which shows that the Nb(3)O₆ octahedron starts to release its distortion with decreasing temperature [see Fig. 6(a)]. As described above, the distortion-released Nb(3)O₆ octahedron may contribute to the transport better than the distorted ones, which will realize rather isotropic conduction in the ab plane. In fact, the anisotropy in resistivity in the ab plane seems to decrease with decreasing temperature as shown in Fig. 3(a). Thus, it seems that electric transport is affected by transforming NbO₆ octahedra to be stable in higher dimensional transport. In other words, dimensional crossover might occur below 150 K. Such dimensional crossover is universally observed in Q1D materials, where the interchain hopping energy is finite. Finite interchain hopping will induce warping of the Fermi surface and effect 2D or 3D behavior below \( t_\perp \) (\( t_\perp \): interchain hopping energy). Naive thinking suggests that the higher dimensionality will induce a metallic state rather than an insulating state.

Finally, we would like to emphasize the unique transport properties in SrNbO₃; a high anisotropy in the thermopower appears at \( T^* \), which is related to the dimensional crossover from a higher dimensional to a Q1D electronic structure associated with the shapes of Nb(1)O₆ and Nb(3)O₆ octahedra. Though Q1D material generally shows a high anisotropy in thermopower, there is no example that exhibits both the low thermopower that is typical for a conventional metal and the high thermopower typical for a semiconductor except for SrNbO₃, as far as we know. Thanks to the enhanced thermopower along the b axis, the \( ZT \) along the b axis is the largest of the \( ZT \)s along all the crystallographic axes. These results show that the Q1D electronic structure enhances the thermopower even in a bulk crystal.

### IV. CONCLUSION

We have systematically investigated the transport properties of a SrNbO₃ single crystal along all crystallographic axes. The resistivity, thermopower, and thermal conductivity were found to be highly anisotropic, which reflects Q1D electronic
and crystal structures. In particular, the thermopower along the $b$ axis is $-170 \mu V/K$, which is 1 order of magnitude higher than the $-15$ and $-25 \mu V/K$ along both the $a$ and the $c$ axes. A high anisotropy in the thermopower appears at $T^*$, which is related to the dimensional crossover from a higher dimensional to a Q1D electronic structure associated with the shapes of Nb(1)O$_6$ and Nb(3)O$_6$ octahedra.

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